

Electronic and Spin States of SrRuO₃ Thin Films: an X-ray Magnetic Circular Dichroism Study

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We report a study of the local magnetism in thin films of SrRuO₃ grown on (111) and (001) oriented SrTiO₃ substrates using x-ray magnetic circular dichroism spectroscopy (XMCD) at the Ru- $L_{2,3}$ edges. The application of the sum rules to the XMCD data gives an almost quenched orbital moment and a spin moment close to the value expected for the low spin state $S = 1$. Full-multiplet cluster calculations indicate that the low spin state is quite stable and suggest that the occurrence of a transition to the high spin state $S = 2$ in strained thin films of SrRuO₃ is unlikely as it would be too expensive in energy.

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Despite being investigated already for about five decades the physical properties of SrRuO₃ keeps fascinating the scientific community. SrRuO₃ is one of the few known 4d transition metal oxide ferromagnets with T_c as high as 160 K^{1,2}. Its non-integer magnetic moment has been interpreted in terms of a surprising rare example of itinerant ferromagnetism in oxides^{3,4}. More recently, the possibility of employing thin films of SrRuO₃ as conducting layer in epitaxial heterostructures of functional oxides has aroused a wide attention from the applied science community⁵.

SrRuO₃ is a perovskite compound with an orthorhombic GdFeO₃ type structure^{6,7}. The orthorhombic distortion arises from the zig-zag tilting, along the c -axis, and rotation, around the b -axis, of the corner-sharing RuO₆ octahedra. Despite this distortion the RuO₆ octahedra remain nearly regular⁷⁻⁹. In a localized picture, the strong crystal field at the octahedral site splits the Ru 4d bands of the Ru⁴⁺ ions into e_g and t_{2g} levels, leading to a low spin (LS) t_{2g}^4 configuration with $S = 1$. Theoretical calculations¹⁰⁻¹³ and a X-ray magnetic circular dichroism (XMCD) study¹⁴ suggest that the orbital moment in SrRuO₃ should be quenched. High magnetic field measurements on a bulk single crystal give a saturated magnetization of $1.6 \mu_B/\text{Ru ion}$ ¹⁵, a value similar to the ordered magnetic moment determined by neutron diffraction experiments^{2,9}.

While the technology for growing high quality SrRuO₃ thin films on (001) oriented SrTiO₃ substrates, SRO/(001)STO, was developed long time ago and is well known⁵, the systematic growth of thin films on (111) oriented SrTiO₃ substrates, SRO/(111)STO, is quite recent^{16,17}. Very surprisingly the first SQUID measurements of SRO/(111)STO films have provided a sat-

urated moment of $3.4 \mu_B/\text{Ru ion}$ ^{17,18}, a value that is much higher than that observed in bulk SrRuO₃ and exceeds the atomic moment of $2 \mu_B/\text{Ru ion}$ expected for a $S = 1$ spin state. In order to explain the SQUID results it has been proposed¹⁸ that the trigonal compressive strain induced by the (111)STO substrate onto the film would stabilize the high spin state (HS) $S = 2$, which is very surprising as a HS state is unusual in 4d oxides. Even more intriguing, an unquenched orbital moment of about $0.32 \mu_B$ has been reported for these strained films on the basis of XMCD measurements at the Ru $M_{2,3}$ edges¹⁸. However, theoretical studies, which investigated the effect of substrate-induced compressive strain on the physical properties of SrRuO₃, could not find evidence in support of the alleged stabilization of a HS state or even suggested the reduction of the magnetic moment from bulk values^{13,19}. Further, a very recent study²⁰ on SRO/(111)STO has reported magnetization values suggesting a LS state, in contradiction with the results published earlier^{17,18}. Understanding the stability of the magnetic ground state of SrRuO₃ is obviously a very important aspect for controlling the magnetic properties of heterostructures involving SrRuO₃ as conducting layer.

In this work we address two questions: 1) whether compressive strain can induce a spin state transition in SRO/(111)STO and 2) whether the orbital moment is quenched. To this end, we have performed an investigation of the XMCD signal at high magnetic field at the Ru $L_{2,3}$ edges of SrRuO₃ films under different compressive strains (trigonal strain for the case of (111)STO substrate and tetragonal strain for the case of (001)STO substrate) compared with the case of a SrRuO₃ single crystal. XMCD is a well-established technique to study local magnetic properties. The XMCD signal can be analyzed by means of sum rules^{21,22}, allowing for a direct

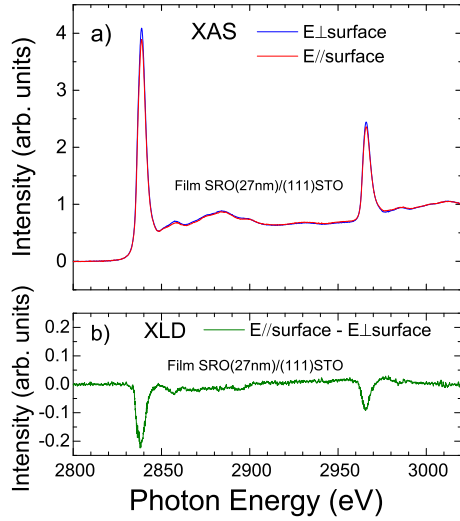


FIG. 1: Ru- $L_{2,3}$ XAS spectra of a 27 nm SRO/(111)STO film for linearly polarized light coming in with the electric field vector \mathbf{E} normal [dark (blue) lines] and parallel [light (red) lines] to the film surface.

experimental determination of the desired quantum numbers L_z and S_z . The energy separation between Ru L_3 and L_2 edges of about 150 eV is much larger than the multiplet effects (a few eV), and therefore the spectra are very suitable also for spin sum rule analysis²². In addition, the signal-to-background ratio at the $L_{2,3}$ edges is higher than at the $M_{2,3}$ edges. We would like to stress that obtaining a reasonable degree of circular polarized light at the photon energies of the L edges of 4d elements is challenging²³ and only thanks to the development of the new BOREAS beamline this XMCD investigation of the Ru $L_{2,3}$ edges has been possible. In addition, a comparison of the line shape to full-multiplet theory can be made to unravel details of the wave functions forming the ground state.

Single crystalline thin films of SrRuO₃ were grown on SrTiO₃ substrates with different orientations by metal-organic aerosol deposition. Thin SrRuO₃ films grown on (001) and (111) oriented substrates were determined by X-ray diffraction (XRD) to have (100)c and (111) orientation, respectively. (In this report, we use pseudocubic notation for SRO films. (110)orthorhombic and (101)orthorhombic is equivalent to (100)c and (111) in the pseudocubic notation). The XRD results show that the films grown on (111) oriented substrates exhibit an elongation of the out-of-plane lattice constant (3.946(1) and 3.950(1) Å for the 80 and 27 nm thick films, respectively) compared to bulk SrRuO₃ (≈ 3.93 Å¹⁸). This systematic evolution of the out-of-plane constant with film thickness (the thinner the film, the larger the out-of-plane constant) is an effect of the strain: under compressive in-plane strain the in-plane lattice constant shrinks, while the out-of-plane lattice constant becomes elongated, in order to roughly preserve the unit cell volume²⁰. The thickness of the SrRuO₃ films was determined by small-

angle X-ray scattering. Details of their preparation and structure characterization are reported in Ref. 24. Large single crystals of SrRuO₃ were grown by floating zone technique. The purity and quality of the crystal were checked by x-ray diffraction. Susceptibility measurements using a MPMS squid magnetometer show a bulk ferromagnetic transition at $T_c = 160$ K for the single crystal, and between 154 K and 147 K for the films depending on the film thickness. The x-ray linear dichroism (XLD) and x-ray magnetic circular dichroism (XMCD) experiments at the Ru- $L_{2,3}$ edges (2800-3000 eV) were performed at the BL29 Boreas beamline at the ALBA synchrotron radiation facility in Barcelona. The energy resolution was 1.4 eV and the degree of circular polarization delivered by the Apple II-type elliptical undulator was adjusted to 70% as balanced trade off between degree of polarization and photon flux required when working at high photon energies and high undulator harmonics. The degree of linear polarization for XLD is close to 100%. The XMCD signal was measured using a magnetic field of 6 Tesla with the sample at a temperature of 50 K. The spectra were recorded using the total electron yield method (by measuring the sample drain current) in a chamber with a vacuum base pressure of 2×10^{-10} mbar. The single crystalline sample was cleaved in situ to obtain a clean sample surface normal to the (110) direction. The XAS spectra were collected in both $B = 6$ T and -6 T applied fields and in groups of four or quartet ($\sigma^+ \sigma^- \sigma^- \sigma^+$ or $\sigma^- \sigma^+ \sigma^+ \sigma^-$, where σ^+ and σ^- indicate photon spin parallel or antiparallel to the applied field, respectively) in order to minimize the effect of any time dependence in the X-ray beam on the measured spectra.

In Fig. 1 we report the Ru- $L_{2,3}$ XAS measured on a 27 nm SRO/(111)STO film for linearly polarized light coming in with the electric field vector \mathbf{E} normal [dark (blue) lines] and parallel [light (red) lines] to the film surface. The Ru 2p core-hole spin-orbit coupling splits the spectrum roughly in two parts, namely the L_3 (at $h\nu \approx 2840$ eV) and L_2 (at $h\nu \approx 2970$ eV) white lines regions. A clear linear dichroism (XLD) can be observed, which is an indication that the film is under in-plane compressive strain. In fact, in-plane compressive strain leads to a trigonal elongation of the RuO₆ octahedron along the (111) axis. As consequence, the t_{2g} orbitals are split in a_{1g} and e_g^π orbitals, with the a_{1g} orbital lying higher in energy and, hence, having more holes. The experimentally observed larger spectral weight for \mathbf{E} normal to the film surface is a result of the uneven hole distribution among the t_{2g} orbitals induced by the strain.

The top panel of Fig. 2 shows the Ru- $L_{2,3}$ XAS measured on SRO/(111)STO and SRO/(001)STO films and, for comparison, on a SrRuO₃ single crystal. The XAS spectra were taken using circular polarized light with the photon spin parallel (σ^+ , red curves) and antiparallel (σ^- , blue curves) aligned to the magnetic field. The difference spectrum ($\sigma^- - \sigma^+$), i.e., the XMCD spectrum, is reported in the bottom panel of Fig. 2. The spectra were collected with the beam

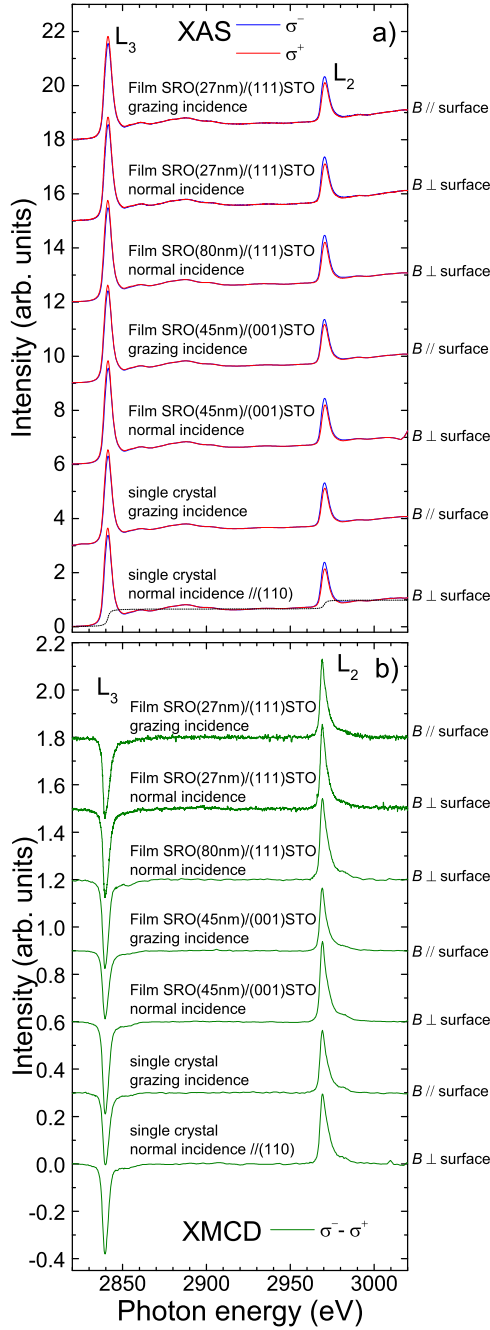


FIG. 2: Ru- $L_{2,3}$ XAS spectra a) and XMCD spectra b) of SrRuO₃ films and bulk single crystal measured at $T = 50$ K and $H = 6$ T. The spectra are vertically shifted for clarity. The spectra taken at normal (90°) and grazing (20°) incidence show a small anisotropy. The dashed black curve represents the edge jump.

in grazing ($B // \text{surface}$) and in normal ($B \perp \text{surface}$) incidence, see Fig. 3 for experimental geometry. The XMCD signal is larger for $B \perp \text{surface}$ than for $B // \text{surface}$ by about 30% for the SRO/(001)STO film, and by about 5% for the 27 nm SRO/(111)STO film. The anisotropy of the XMCD signal agrees with the picture

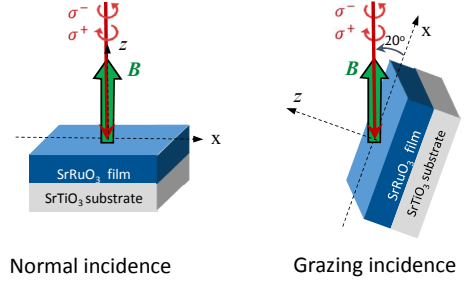


FIG. 3: Setup of the XMCD experiments: the magnetic field B is applied parallel to the Poynting vector of the circularly polarized photons and forms an angle of 90° (20°) in normal (grazing) incidence with the sample surface.

of an out-of-plane easy axis for SrRuO₃ films grown on STO as reported in literature^{17,20}. The reduced magnetic anisotropy shown by our XMCD measurements in the case of SRO/(111)STO films with respect to SRO/(001)STO film is in fair agreement with previous SQUID measurements²⁰. Both XAS and XMCD spectra measured on the SRO/(111)STO and SRO/(001)STO films appear fairly identical to those measured on the bulk single crystal, without clear evidence of changes in the spectral lineshape and in the size of the XMCD signal that otherwise could suggest a different spin state.

Sample	incidence	$L_z/2S_z$	L_z	$2S_z$
Crystal	$B \perp \text{surface}$	0.01	0.01	1.9
Crystal	$B // \text{surface}$	0.01	0.02	1.7
SRO(45nm)/(001)STO	$B \perp \text{surface}$	0.01	0.01	1.9
SRO(45nm)/(001)STO	$B // \text{surface}$	0.01	0.02	1.5
SRO(80nm)/(111)STO	$B \perp \text{surface}$	0.00	0.00	2.0
SRO(27nm)/(111)STO	$B \perp \text{surface}$	0.01	0.03	1.9
SRO(27nm)/(111)STO	$B // \text{surface}$	0.01	0.02	1.8

TABLE I: $L/2S$ ratio, orbital and spin moment as estimated using sum rules. The values were divided by a factor 0.7 to take into account that the beam was only 70% circular polarized.

The material metallicity and life time broadening (~ 2 eV) may limit the information that can be obtained from the lineshape about the Ru ground state. However, it is possible to use the sum rules for XMCD developed by Thole and Carra et al.^{21,22} to extract from our XMCD data the orbital (L_z) and spin ($2S_z$) moments:

$$L_z = \frac{4}{3} \cdot \frac{\int_{L_{2,3}} (\sigma^+ - \sigma^-) dE}{\int_{L_{2,3}} (\sigma^+ + \sigma^-) dE} \cdot N_h, \quad (1)$$

$$2S_z + 7T_z = 2 \cdot \frac{\int_{L_3} (\sigma^+ - \sigma^-) dE - 2 \int_{L_2} (\sigma^+ - \sigma^-) dE}{\int_{L_{2,3}} (\sigma^+ + \sigma^-) dE} \cdot N_h, \quad (2)$$

For ions in octahedral symmetry the magnetic dipole moment T_z is a small number and can be neglected compared to S_z ²⁵. The number of holes in the 4d shell was

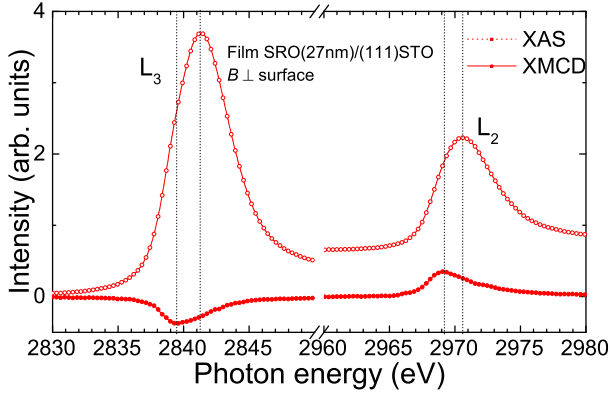


FIG. 4: Zoom of Ru- $L_{2,3}$ isotropic XAS and XMCD spectra of a 27 nm thin film of SrRuO₃ grown on (111) oriented SrTiO₃ substrate. The vertical dashed lines corresponds to the energy position of the maxima in the XAS and XMCD spectra.

estimated to be about $N_h = 5.2$ by our cluster calculations, in agreement with previous estimates²⁶, reflecting the highly mixed $p-d$ covalency of the ground state in SrRuO₃. In estimating the XAS intensities, the edge jump background, described as arctan function, has been subtracted from the XAS spectra (dashed curve in Fig. 2). The results of the application of the sum rules are reported in table 1. The orbital moment L_z is found to be almost quenched for all samples, including the SRO/(111)STO. The spin contribution to the magnetic moment in the 80 and 27 nm SRO/(111)STO films in normal incidence is found to be close to the value expected for a $S = 1$ spin state and very similar to that found for the bulk single crystal²⁷. These results are in clear contradiction with the much larger saturated moment values reported earlier¹⁸ from SQUID measurements on SRO/(111)STO films.

As mentioned before, the spectra seem to be rather featureless, but a closer look reveals that for both L_3 and L_2 edges the maximum in intensity of the XAS spectrum lies 1.5 eV higher in energy position than that of the XMCD spectrum (see Fig. 4). We see exactly the same difference in energy position in the spectra of all samples. A similar energy shift of the XMCD peak with respect to the XAS peak was previously observed for the Ru- $M_{2,3}$ edges¹⁴ and can be understood considering that only the t_{2g} orbitals contribute to the XMCD signal, while both t_{2g} and e_g orbitals contribute to the XAS spectrum with the XAS maximum corresponding to the signal from the unoccupied e_g levels. Therefore, this energy position difference provides a very important information as it reflects the crystal field splitting $10Dq$ between the t_{2g} and e_g orbitals. In order to determine quantitatively $10Dq$ we have performed simulations of the XAS and XMCD spectra using the well-proven full-multiplet configuration-interaction approach^{28,29}. It accounts for the intra-atomic $4d-4d$ and $2p-4d$ Coulomb interactions, the atomic $2p$ and $4d$ spin-orbit couplings, the oxygen $2p-4d$ hybridization, and local crystal field

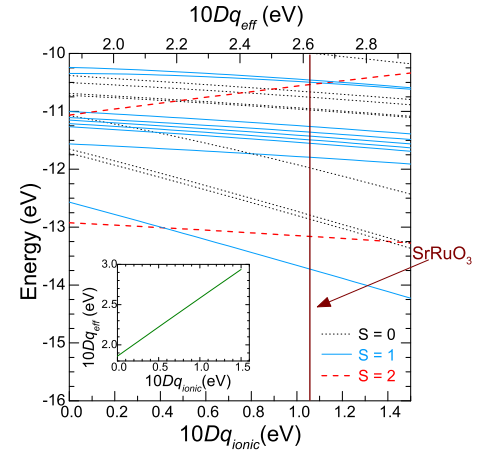


FIG. 5: Energy level diagram of the Ru⁴⁺ ion as a function of the ionic crystal field $10Dq$ in a cubic local symmetry. Black dotted, blue solid and red dashed lines correspond to levels with $S = 0, 1$ and 2 spin state, respectively. The inset shows the evolution of the effective crystal field versus ionic crystal field. The vertical magenta solid line indicates the $10Dq_{eff}$ of SrRuO₃ as obtained by the simulation of the XAS and XMCD spectra.

parameters. In the simulations we considered a RuO₆ cluster with a cubic symmetry as the octahedra in bulk SrRuO₃ are fairly regular⁷⁻⁹. The calculations were performed using the XTLS 8.3 code³⁰ with the parameters given in Ref. 31. We applied to the spectra additional Gaussian (1.4 eV) and Lorentzian (2 eV) broadening in order to take into account experimental resolution and lifetime effects, respectively.

In Fig. 5 we report the energy level diagram of the Ru⁴⁺ ion as a function of the ionic crystal electric field $10Dq_{ionic}$ in a cubic local symmetry. The energy difference between the two $S = 2$ levels with orbital occupation $t_{2g}^3 e_g^1$ and $t_{2g}^2 e_g^2$ (bottom and top red dashed line in Fig. 5, respectively) can be taken as a measure of the effective crystal electric field $10Dq_{eff}$, i.e. the splitting between t_{2g} and e_g levels including the effect of the hybridization with the oxygens. The diagram shows that for $10Dq_{eff} > 2.15$ eV ($10Dq_{ionic} > 0.41$ eV) the level with configuration $t_{2g}^4 e_g^0$ is the lowest energy level (bottom solid blue line) and the ground state of Ru⁴⁺ ion has $S = 1$ spin state. As the crystal field is reduced across the critical value of $10Dq_{eff} \leq 2.15$ eV ($10Dq_{ionic} \leq 0.41$ eV) the $t_{2g}^3 e_g^1$ level (red dashed line) becomes the lowest energy level and the HS state is stabilized. The non-magnetic $S = 0$ state (black dotted lines) lies always much higher in energy and never becomes the ground state for any value of the cubic crystal field.

In Fig. 6 we show the comparison of the simulated XAS and XMCD spectra at the $L_{2,3}$ edges with the experimental spectra measured on SRO/(111)STO film. The simulated XAS and XMCD spectra were calculated for different values of the ionic and effective crystal field splitting. For the sake of clarity, the spectra were normalized to the height of the peak and the XMCD signal at the

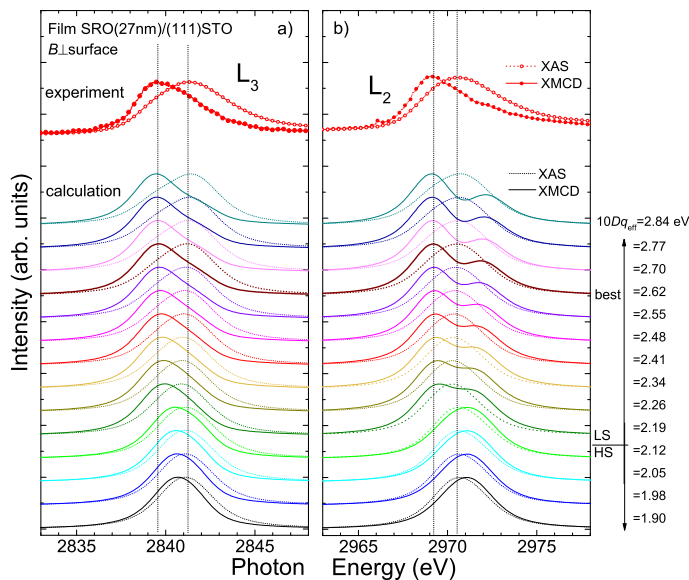


FIG. 6: Experimental (top) and simulated (below) XMCD and XAS spectra for different $10Dq_{eff}$ values at the L_3 (panel a) and L_2 (panel b) edges. For a better comparison the spectra were normalized to the peak height and the Ru L_3 XMCD spectra was reversed. The vertical dotted lines correspond to the energy position of the maxima in the XAS and XMCD spectra.

L_3 edge was reversed. The calculated spectra show that the peak position depends on the value of crystal field splitting. The experimental energy separation between the maxima of the XMCD and XAS spectra can be correctly simulated for $10Dq_{eff} = 2.62$ eV and the lineshape of the calculated spectra is fairly similar to that of the experimental spectra. For such a value of $10Dq_{eff}$ the Ru^{4+} ions are in a LS $S = 1$ ground state. The HS spin

state becomes stable only for smaller crystal field splitting, $10Dq_{eff} < 2.15$ eV. In the hypothesis of a HS spin state as a ground state the simulated XMCD spectrum looks very different from the experimental one: 1) at the L_3 and L_2 edge the XMCD lineshape is not anymore asymmetric; 2) at the L_2 edge the XMCD maximum occurs at higher photon energy than the XAS maximum, which is opposite to what has been experimentally observed. As it can be seen in the energy level diagram reported in Fig. 5 $SrRuO_3$ is located very far from the stability region for the HS state.

To summarize, we have used XMCD spectroscopy to investigate the local magnetism in thin films of $SrRuO_3$ grown on (111) and (001) oriented $SrTiO_3$ substrates. We have found that the orbital moment is almost quenched and the spin is close to the value expected for a $S = 1$ spin state. From a comparison of the experimental with simulated spectra we could determine the effective crystal field. The hypothesis of a compressive strain-induced spin state transition, as proposed in literature on the basis of SQUID measurements, can be ruled out as the stabilization of the high spin state with $S = 2$ would be too costly in energy.

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 - ²⁷ The spin contribution obtained by Okamoto et al.¹⁴ from Ru- $M_{2,3}$ XMCD measurements is only $\sim 0.6 \mu_B$. However, as the authors admitted, the electron-core-hole interaction, which in the case of the $M_{2,3}$ edges mixes $p_{3/2}$ and $p_{1/2}$ components, the use of scraped surfaces of polycrystalline samples and the relative small magnetic field (2 T) might have caused an underestimation of the moment reported in the latter work.
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